# Surface Interactions in the Pt/<sub>Y</sub>-AI<sub>2</sub>O<sub>3</sub> System

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Using temperature-programmed reduction, Pt oxide deposited on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> was found to be present in two phases: a particulate phase in which the Pt oxide is aggregated and has little interaction with the support, and a dispersed phase in which Pt oxide has a larger interaction with the support. The lower the concentration of Pt oxide in the dispersed phase, the stronger is the Pt oxide-support interaction. Using selective chemisorption of H<sub>2</sub> and CO on Pt/ $\gamma$ -Al<sub>2</sub>O<sub>s</sub> samples, the Pt oxide concentration in the dispersed phase was found to increase proportionally with Pt loading until a saturation concentration of 2.2  $\mu$ mol Pt/m<sup>2</sup> (BET) is reached. Beyond this concentration, the excess Pt oxide aggregates to form a particulate phase. At 50O°C in  $O<sub>2</sub>$  a quasi-equilibrium between Pt oxide in the dispersed phase and that in the particulate phase is reached. The behavior of Pt oxide on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> during other various heat treatments was found, using transmission electron microscopy, to depend on Pt loading as well as temperature and chemical atmosphere.

### INTRODUCTION

A number of previous studies *(1-6)* of platinum oxide deposited on a  $\gamma$ -alumina support have shown the presence of two platinum oxide phases: a two-dimensional dispersed phase, in which Pt oxide is present in  $Pt-Al_2O_8$ -Cl or  $PtO_2-Al_2O_3$  complexes *(1-3)* and is soluble in acids and acetylacetone, and a three-dimensional particulate phase in which Pt oxide is present in erystallites and is insoluble in acids. The existence of two phases in other supported transition metals and metal oxides has also been reported *(7-16).* It has been demonstrated that at low concentration, transition metal oxides on oxide supports can be dispersed into a two-dimensional phase by heat treatment in air or vacuum at about 500°C *(12-15).* At high metal concentration, a saturated dispersed phase is in equilibrium with a particulate phase after this treatment. The saturation concentration of the dispersed phase varies with the metal oxide and also with the composition of the support.

This work continues the investigation of the two phases of Pt oxide on  $\gamma$ -alumina with the following objectives: (i) to give further evidence of the existence of two phases, aside from the solubility differences, (ii) to determine the saturation concentration of the dispersed phase, and (iii) to describe the behavior of Pt on  $\gamma$ -alumina during heat treatment in oxidizing and reducing atmospheres with reference to the two phases.

#### EXPERIMENTAL

### *I. Materials*

A granular  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support was made by agglomerating Dispal-M  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> powder (Conoco Chemicals) by wetting with water, drying, and calcining at 600°C. The result-





<sup>a</sup> Fresh Pt/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> samples were calcined at 400°C in air and reduced at 300°C by H<sub>2</sub> before adsorption.

ing solid mass was crushed and sieved and the 0.5- to 1.0-mm-diameter fraction was chosen for the support. An adsorption-impregnation method was used to prepare the samples (Table 1). The support for each sample was immersed in distilled water into which aqueous  $H_2PtCl_6$  of a desired concentration was added. The samples were evaporated and dried in air at 120°C. Samples with high Pt loading were rewetted with water and redried to distribute the Pt ions uniformly. Each sample was calcined in air at 400°C for 2 hr. The Pt content of each sample was determined by X-ray fluorescence. The chloride content of some selected samples was determined by the same method.

Research grade  $N_2$  and CO were used without further purification. Hydrogen was purified by passage over an Engelhard Deoxo Pd catalyst and a 4-A molecular sieve at 78° K. PtO<sub>2</sub> (40 mesh, Ventron Corp.) was used without further purification. This  $P<sub>tO</sub>$  was found to be insoluble in an HF solution. An analysis by X-ray diffraction indicated the presence of a small amount of metallic Pt.

### *II. Adsorption Measurements*

The apparatus and procedure for measuring BET area, and hydrogen and carbon monoxide adsorptions of supported platinum samples, are described in previous papers *(14, 15).* All samples were pretreated in  $H_2$  at 300°C (in some cases at 500°C) for 2 hr and outgassed at the same temperature for another hour.

# *IlL Temperature-Programmed Reduction (TPR)*

TPR measurements were made in a flow system similar to that described by Robertson *et al.* (17). A small amount of  $P<sub>t0<sub>2</sub></sub>$  or  $Pt/\gamma$ -Al<sub>2</sub>O<sub>3</sub> was precalcined in air at  $500^{\circ}$ C and placed in a TPR cell at  $-75^{\circ}$ C through which  $H_2-Ar(15:85)$  gas flowed. The temperature of the sample was programmed to rise at a constant rate of 8°C/ min and the amount of  $H_2$  uptake during the reduction was measured by a thermal conductivity detector.

# *I V. Transmission Electron Microscope (TEM)*

TEM micrographs were obtained using a Siemens Elmiskop I TEM. The selected area electron diffraction patterns and energy dispersive X-ray analyses were obtained using a Philips 300 TEM. The extraction replication procedure used in making these measurements is similar to that described by Benesi *et al. (18)* and Dalmai-Imelik *et al.* (19). A small amount of ground sample was spread on a microscope slide and was coated with a layer of evaporated carbon. The carbon film containing sample

particles was separated from the slide by floating it first on the surface of a  $10\%$  HF solution to dissolve the alumina completely, and then on the surface of distilled water to wash away the HF. The replica film was then picked up from the water surface using microscope grids and allowed to dry on bibulous paper before observation in the TEM.

### RESULTS

### *I. Chemisorption Measurements*

*A. Fresh samples~* Adsorption isotherms of  $H_2$  and CO on six fresh  $Pt/\gamma - Al_2O_3$ samples of varying Pt loading were measured at  $25^{\circ}$ C. H<sub>2</sub> and CO uptakes of each sample were determined from these measurements by extrapolating the isotherms to zero pressure of the adsorbate. These uptakes along with H/Pt and CO/Pt for each sample, the ratios of H and CO adsorbed at 25°C to the total number of Pt atoms in the sample, are summarized in Table 1.



FIG. 1. Chemisorption of  $H(A)$  and  $CO$  (B) on  $Pt/\gamma$ -Al<sub>2</sub>O<sub>3</sub> samples as a function of Pt loading: (O) heated 2 hr in H<sub>2</sub> at 300°C; ( $\triangle$ ) heated 2 hr in H<sub>2</sub> at 500°C; ( $\triangle$ ) heated 2 hr in H<sub>2</sub> at 750°C; ( $\diamond$ ) heated 5 hr in  $O_2$  at 500°C.



FIG. 2. CO chemisorption on  $Pt/\gamma$ -Al<sub>2</sub>O<sub>3</sub> samples after 3-hr heat treatments at various temperatures: (A) sample 6 *in vacuo;* (B) sample 6 in dried air; (C) sample 3 in dried air.

It has been suggested that the stoichiometric H/Pt and CO/Pt ratios for a completely dispersed sample are unity *(20)*  and 0.87 *(21),* respectively. So the high ratios of  $H/Pt$  (1.18-1.26) and  $CO/Pt$  $(0.83-0.85)$  in the samples of low Pt loading (samples 1, 2, and 3) probably indicate complete dispersion of Pt in these samples. H/Pt ratios higher than unity have been found previously on well-dispersed supported Pt samples  $(6, 22-25)$  and were attributed to multiple adsorption *(23, 25)*  and adsorption on the interstices of monolayer Pt atoms  $(6)$ . On the other hand, the H/Pt and CO/Pt ratios in the samples of high Pt loading (samples 4, 5, and 6) are much lower than unity and 0.87, respectively. These ratios decrease sharply with increasing Pt loading.

*B. Heat treatments in oxidizing and reducing atmospheres.* A comparison was made in Fig. 1 of H and CO uptake on samples 1 to 6 after each step in the following sequence of heat treatments: (i)  $H_2$  at  $300^{\circ}$ C for 2 hr (points 0), (ii)  $H_2$  at 500 or 750°C (points  $\triangle$  or  $\triangle$ ) for 2 hr, and (iii)  $O_2$  at 500°C for 5 hr (points  $\Diamond$ ). The decrease in accessible Pt area in all samples following heat treatment in  $H_2$  at 500 or 750°C indicates sintering. The general in-



FIG. 3. Temperature-programmed reduction of PtO<sub>2</sub> and Pt/ $\gamma$ -Al<sub>2</sub>O<sub>3</sub> samples: (A) PtO<sub>2</sub> powder; (B) sample 6; (C) sample 5; (D) sample 4; (E) sample 2; (F) sample 1 (see Table 2).

crease in accessible area following heat treatment in  $O_2$  at 500°C indicates redispersion of these  $H_2$ -sintered samples.

Another comparison was made in Fig. 2 of the degree of sintering in flowing dried air of samples 3 (curve C) and 6 (curve B) in the temperature range 400 to 750°C. Sintering data for sample 6 in vacuum (curve A) was included as a reference. Sample 6 sinters substantially between 400 and 600°C unlike sample 3 which sinters negligibly. At temperatures higher than 600°C, sintering in the two samples is almost identical.

### *II. Temperature-Programmed Reduction*

In this study a nonsupported  $P<sub>to</sub>$  sample and five  $\gamma$ -alumina-supported samples of varying Pt loading were reduced at temperatures ranging from  $-75$  to  $500^{\circ}$ C. The results shown in Fig. 3 and Table 2 indicate that the nonsupported  $PtO<sub>2</sub>$  was reduced below 25 $^{\circ}$ C with a maximum rate at  $\sim$ 5 $^{\circ}$ C (curve A). The  $10\%$  difference between  $calculated$  and observed values of  $H<sub>2</sub>$ uptake (Table 2) for this sample is probably due to the presence of the small amount of metallic Pt mentioned previously. In the samples of high Pt loading (curves B, C, and D), reduction occurs in two temperature ranges: 5 to 20°C and  $\sim 90^{\circ}$ C. In the samples of low Pt loading (curves E and F), only the high temperature reduction was found.

### *I[I. Particle Size Measurements*

TEM can be used to directly measure the particle size of Pt supported on a very thin alumina film *(26).* When granular silica or alumina is used as support material, interference from support particles may make it impossible for direct TEM measurement and very difficult for measurement with extraction replicas *(27, 28).*  However, a carbon replica in which the support particles were dissolved with a dilute HF solution allowed Pt/silica and Pt/alumina samples to be successfully measured by Benesi *et al. (18)* and Dalmai-Imelik *et al. (19).* A similar method is used in this study to measure alumina-supported Pt oxide particles which are insoluble in HF solution.

Four sets of transmission electron micrographs were made in which four samples of varying Pt loading were subjected to different sequential heat treatments. Pt particle sizes were measured directly from the micrographs at a magnification of

TABLE 2

Temperature-Programmed Reduction of Platinum Oxide on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub>

Sample	Pt in sample $(\mu \text{mol})$	Observed H <sub>2</sub> uptake $(\mu \text{mol})$	Cal- culated <sup>«</sup> H <sub>2</sub> uptake $(\mu \text{mol})$	$T$ max rate $(^{\circ}C)$
$P$ t $O_2$ powder	17.5	32.6	35.0	5
6	73.0	36.4	146.0	10 90
5	22.9	38.5	45.8	20 100
4	11.5	21.0	23.0	175
2	13.9	27.6	27.8	220
1	14.1	24.4	28.2	245

a **Based on the assumption that** all Pt is **present as** PtO2.

Sample	Heat treatment sequences	Particle size range (TEM) $(\AA)$	Average particle size (TEM) $(\AA)$	Average particle size (electron diffraction line broadening) $(\AA)$
1	$500^{\circ}$ C, 3 hr, $O_2$		$-a$	
1	300 °C, 3 hr, $H_2$		$-a$	
1	$500^{\circ}$ C, 3 hr, H <sub>2</sub>	$10 - 30$	15	
1	$750^{\circ}$ C, 3 hr, $H_2$	$10 - 30$	15	
3	$500^{\circ}$ C, 3 hr, H <sub>2</sub>	$10 - 35$	20	
3	$500^{\circ}$ C, 5 hr, O <sub>2</sub>	$10 - 35$	20 <sup>b</sup>	
$\mathbf{3}$	600°C, 4 hr, $O_2$	$10 - 75$	35	
3	$750^{\circ}$ C, 4 hr, $O_2$	200-800	300	
$6\phantom{1}$	$500^{\circ}$ C, 3 hr, $O_2$	$10 - 30$	15	13
6	$300^{\circ}$ C, 3 hr, $H_2$	$10 - 55$	30	
6	$500^{\circ}$ C, 3 hr, H <sub>2</sub>	$10 - 55$	30	
6	$750^{\circ}$ C, 3 hr, $H_2$	$10 - 30$	20	18
		$55 - 150$	70	
6	$750^{\circ}$ C, 5 hr, H <sub>2</sub>	$10 - 30$	20	
		$60 - 300$	130	
6	500°C, 16 hr, $O_2$	60-300	130 <sup>c</sup>	

TABLE 3 The TEM Data on Particle Size

<sup>a</sup> No particles observed.

 $b$  Less than  $5\%$  of the particles were left after this heat treatment.

Used as standard for line broadening in electron diffraction.

550,000X or 350,000X. For some of the samples, electron diffraction patterns were also taken and particle sizes were calculated from measurements obtained from microdensitometer traces of the diffraction patterns using the Sherrer line broadening equation. These data are summarized in Table 3. Figure 4 shows representative areas of selected micrographs.

After heating sample 1 in  $O_2$  at  $500^{\circ}$ C and after heating in  $H_2$  at 300°C, no particles were found in the micrographs (Table 3) and no Pt was detected on the extraction replicas, indicating that this Pt was soluble in the HF solution. Only after heat treatment of sample 6 in  $H<sub>2</sub>$  or sample 3 in air at 750°C was a large increase in particle size found. It is interesting to note that heat treatment in  $H_2$  at 750°C can produce a bimodal size distribution (Fig. 4C and Table 3). Only the small particles are redispersed during the subsequent heat treatment in  $O_2$ ; the larger crystallites remain intact (Fig. 4D).

#### DISCUSSION

# *I. The Existence of Two Phases and the Dispersion Limit*

The temperature-programmed reduction gives evidence of two phases of Pt oxide on a  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> support, in addition to evidence from solubility differences mentioned previously. Samples 4 to 6 (curves B, C, and D, Fig. 3) show both phases : a particulate phase where the Pt oxide has little interaction with the alumina support as seen by the reduction at  $\sim 10^{\circ}$ C, resembling the reduction of unsupported bulk  $PtO<sub>2</sub>$ ; and a dispersed phase where the Pt oxide interacts more strongly with the support, and thus is reduced at higher temperatures.







TABLE 4 Chloride Content in Sample 4 after Sequential Treatments

Sequential treatments	$Cl^-$ content (mg/g)	
Dried at 110°C, 5 hr, air	58.0	
$400^{\circ}$ C, 5 hr, flowing air	24.5	
300°C, 3 hr, flowing $H_2$	11.5	
$500^{\circ}$ C, 5 hr, flowing H <sub>2</sub>	6.5	
500°C, 20 hr, flowing air	< 1.0	

In the samples of low Pt loading (curves E and F), only dispersed phase Pt oxide is found. The  $H_2$  uptake measurements during these reductions (Table 2) indicate that Pt oxide in the dispersed phase is  $P<sub>t</sub>O<sub>2</sub>$  while Pt in the particulate phase is probably mixed  $PtO<sub>2</sub>$ ,  $PtO$ , and metallic Pt. The large increase of the reduction temperature with dilution of  $P<sub>t</sub>O<sub>2</sub>$  in the dispersed phase indicates that all the dispersed  $PtO<sub>2</sub>$  molecules are in close contact with the support and they are probably separated from each other. The larger this separation, the stronger is the  $PtO<sub>2</sub>$  $Al<sub>2</sub>O<sub>3</sub>$  interaction. This interaction may result in a dissolution of  $PtO<sub>2</sub>$  into the surface layer of  $Al_2O_3$  or in the formation of a  $PtO_2 \cdot Al_2O_3$  complex as previously suggested  $(1-3)$ . The PtO<sub>2</sub> in the dispersed phase would consequently be soluble in the acetylacetone and dilute HF solution. As pointed out previously (3, *29)* chloride is not required for the formation of this soluble stable complex. The chloride ions in the support of the samples used in this study (Table 4) may stabilize Pt ions on alumina (1) and thereby increase their dispersibility.

Although the two phases are strictly defined only for Pt oxide on alumina, the chemisorption data (Table 1) shows that after reduction in  $H_2$  at 300°C, samples 1 to 3 remain completely dispersed. In these samples of low Pt loading, Pt area increases proportionally with Pt loading (Fig. 1) until it reaches a dispersion limit of about 2.2  $\mu$ mol Pt/m<sup>2</sup> (BET). Beyond this limit the excess Pt oxide aggregates to form particles or crystallites.

# *II. The Quasi-equilibrium of Pt Oxide on*   $\gamma$ - $Al_2O_3$  in  $O_2$  at  $500^{\circ}$ C

The surface concentration of exposed Pt atoms in fresh samples of high Pt loading (reduced at 300 $^{\circ}$ C in H<sub>2</sub>) can exceed the dispersion limit of 2.2  $\mu$ mol Pt/m<sup>2</sup> (BET). However, heat treatment in  $O_2$  at  $500^{\circ}$ C decreases this surface concentration to the dispersion limit (Figs. 1 and 2). Heat treatment of these samples in flowing  $H_2$  at 500°C decreases the exposed Pt surface concentration below the dispersion limit, but heat treatment in  $O_2$  at 500 °C increases it back to the dispersion limit (Fig. 1). These results indicate a saturation concentration of Pt oxide in the dispersed phase and a quasi-equilibrium between Pt oxide in the dispersed phase and that in the particulate phase at  $500^{\circ}$ C in  $O_2$ . The saturation concentration of dispersed phase Pt oxide is  $\sim$  2.2  $\mu$ mol/m<sup>2</sup> (BET). Pt oxide at this concentration, if closely packed in a monolayer, can cover only about 15% of the surface area of the alumina support. The crystallite size of Pt oxide in the particulate phase at equilibrium is  $\sim$ 15 Å (Table 3, sample 6 after 500°C heat treatment in  $O_2$ ).

A similar quasi-equilibrium was also found in other supported transition metal oxide systems *(13-15).* Saturation concentration was found to vary mainly with support material and to a lesser extent with transition metal. For example, the saturation concentration of cobalt oxide on zirconia was found to be  $\sim 8 \mu \text{mol}$  Co/m<sup>2</sup> (BET) *(13)* while in Re and Rh oxide on  $\gamma$ -Al<sub>2</sub>O<sub>3</sub> it was 2 to 2.5  $\mu$ mol/m<sup>2</sup> (BET) *(14, 15).* 

# *III. Behavior of the Two Phases in Oxidizing and Reducing Atmospheres*

The behavior of supported Pt samples during heat treatment in oxidizing and

reducing atmospheres depends on (i) Pt loading, (ii) state of dispersion, and (iii) temperature of heat treatment. Behavior in  $O_2$  at 500°C (Figs. 1 and 2 and Table 3) can be interpreted with reference to the equilibrium of Pt oxide in the two phases. For samples of Pt loading below the saturation concentration (samples 1-3), all the Pt oxide can be dispersed during heat treatment at  $500^{\circ}$ C in  $O_2$ . Fresh samples of Pt loading above the saturation concentration sinter during heat treatment at  $500^{\circ}$ C in  $O_2$  to reach the saturation concentration of the dispersed phase so the two Pt oxide phases will be in equilibrium.

Heat treatment in hydrogen at various temperatures and in  $O_2$  or air at  $\geq 600^{\circ}C$ reduces or decomposes Pt oxide to form metallic Pt which aggregates (Table 3). In  $H_2$  at 300°C, Pt atoms formed from the dispersed phase  $PtO<sub>2</sub>$  remain completely dispersed. It is uncertain whether these Pt atoms are aggregated to form discontinuous monolayer patches (6) or small clusters of about 6 atoms *(30),* or remain atomically dispersed *(31).* On the other hand these same Pt atoms after reduction at 500 or 750 $\mathrm{^{\circ}C}$  by H<sub>2</sub> aggregate to form particles of 10 to 35 A and their accessibility to  $H_2$  and CO adsorption decreases. Particulate phase  $PtO<sub>2</sub>$  crystallites grow somewhat larger in  $H_2$  even at 300°C. At 750°C Pt particles in two size ranges are produced in samples of Pt loading above the saturation concentration. Only the small particles of 10 to 30 A, similar to those formed from dispersed phase  $P<sub>t</sub>O<sub>2</sub>$ under these conditions, are redispersed in  $O<sub>2</sub>$  at 500°C. Large particles also form in  $O_2$  at  $\geq 600^{\circ}$ C, even in samples with Pt loading below the saturation concentration. At 750°C, the aggregation of Pt particles in  $O<sub>2</sub>$  is much faster than it is in  $H<sub>2</sub>$ .

Loss of Pt area during heat treatment in  $H<sub>2</sub>$  at 500°C and its subsequent recovery after heat treatment in  $O_2$  at 500°C were observed previously *(32-34).* Dautzenberg *et al.* attributed the loss to the reduction of the alumina support during heat treatment in H<sub>2</sub> at 500°C to form a Pt-Al<sub>2</sub>O<sub>3-x</sub> complex or a Pt3A1 alloy which does not adsorb hydrogen and from which Pt can be regenerated by heat treatment in  $O<sub>2</sub>$  at 500°C. The above TPR results do not support this interpretation. Reduction of Pt oxide on alumina is complete at 300°C and no further reduction is found at 500°C in  $H<sub>2</sub>$  (Fig. 3).

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